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Corresponding Author:	Manoj Mukund Kothawale, MSc, SET, Ph.D. Sci and Commerce Mapusa, Goa INDIA		
Corresponding Author Secondary Information:			
Corresponding Author's Institution:	Sci and Commerce		
Corresponding Author's Secondary Institution:			
First Author:	Manoj Mukund Kothawale, MSc, SET, Ph.D.		
First Author Secondary Information:			
Order of Authors:	Manoj Mukund Kothawale, MSc, SET, Ph.D.		
	Manoj Mukund Kothawale, MSc, SET, Ph.D.		
	Rajesh Pednekar, M.Sc, SET, Ph.D.		
	Sher Singh Meena, M.Sc		
Order of Authors Secondary Information:			

Synthesis of Superparamagnetic Nanoparticle Ni_{0.50}Zn_{0.50}Fe₂O₄ using Wet Chemical Method

Manoj M. Kothawale.¹, Rajesh Pednekar .², Sher Singh Meena .³

¹Department of Physics, D.M'S. College of Arts Science and Commerce, Assagao, Goa 403 507, India

²Department of Chemistry, DM'S College of Arts, Science and Commerce, Assagao Bardez Goa, 403 507, INDIA

³Solid State Physics Division, Physics Division, Bhabha Atomic Research Centre, Mumbai, 400 085, INDIA

Tel: 91-9421154390

Email: <u>manojkothawale@yahoo.com;</u>

Abstract. nanoparticles The of $Ni_{0.50}Zn_{0.50}Fe_2O_4$ were prepared using cotton base wet chemical method. The XRD spectrum show single phase cubic spinel structure and crystallize size calculated using Debye Scherrer formula for 311 planes was found to be 41 nm \pm 5%. The nano range sizes of particle were confirmed using TEM and AFM images. The porosity value of the sample is promising towards sensor applications. Infrared Spectroscopic results reveals two main absorption bands indicating sample is having single phase spinel structure with two sub lattices. The high DC resistivity of the sample is attributed to nano range particle sizes. The SPM nature was confirmed from variation of magnetic moment with applied magnetic field

using VSM. The Mössbauer spectrum recorded at room temperature show two major sextets corresponding to A-site and B-site Fe ions. The relaxed sextet is associated with the atoms near the of surface the nano particles disordered spins. experiencing The central doublet in Mössbauer spectra of the sample indicates superparamagnetic (SPM) particles. The isomer shift results show all the Fe ions are in high state.

Keywords: Ni–Zn ferrites. XRD. TEM. Superparamgnetism. isomer shift. DC resistivity.

1. Introduction

Spinel ferrites are technologically important for several electromagnetic devices because of their high Curie temperature, high permeability, and high

electrical resistivity, low eddy current and low dielectric loss. Ni-Zn ferrites are commercially important due to their uses in devices operating at high frequency such as radio frequency coils, transformer cores, etc. [1]. The properties of ferrite are sensitive to method of preparation, compositions, particle sizes and size distribution [2]. In the present work, nanoparticles of Ni_{0.50}Zn_{0.50}Fe₂O₄ were prepared using cotton base wet chemical method. Properties such as particle sizes, saturation magnetization, DC electrical resistivity. porosity and Mossbauer hyperfine parameters were studied.

2. Experimental

Calculated amount of ferric nitrate, nickel nitrate, and zinc nitrate (all salts of AR grade) in stoichiometry proportions were dissolved in minimum amount of distilled water. This solution was soaked using optimum amount of cotton. The soaked cotton pieces were placed in crucible and heated in conventional box type furnace for 6 hr at 600°C. The decomposition of soaked solution resulted into fine powder of Ni-Zn ferrite. The powder obtained was used for characterization. The X-ray powder diffraction pattern was recorded on Rigaku X-ray diffractometer using CuK α radiation and 2 θ scanning range from 20° to 80°. IR spectrum of sample was recorded using Shimatzu Fourier transform infrared (FTIR) 8900 spectrometer in the range of 300 cm^{-1} to 4000 cm^{-1} . The pallet was prepared with KBr to sample ratio as 1:100 for IR measurements. Transmission electron microscope (TEM) image was recorded Hitachi H7500. Atomic force on ficroscope (AFM) image was recorded on NT-MDT's Integra AFM via semi contact mode. Variation of magnetic moment at room temperature was recorded using VSM (Lakeshore model 7404) up to field of 2Tand corresponding hysteresis curve was obtained. The room temperature Mössbauer spectrum of the sample was recorded in constant acceleration mode using a ⁵⁷Co. The calibration of the velocity scale was done using ⁵⁷Fe metal foil. Mass density was calculated using Archimedes principle [3]. The freshly prepared powder was pressed into pellet (10 mm dia and 2 mm thick) by applying pressure of 75 KN for 5 min. The pellet was then coated with silver on both surfaces for having good electrical contact. The DC resistivity was

measured by two probe method using Keithley electrometer in the temperature range of 30 °C to 500 °C in steps of 5°C.

3. Results and discussion

3.1 XRD, TEM and AFM analysis

The X-ray diffraction (XRD) pattern of the sample is as shown in Fig. 1. The strongest refection from (311) plane denotes the spinel phase. The positions of peaks comply with reported JCPDS file no 8-234. This confirms that sample under study is having single phase cubic spinel structure. The observed lattice constant value of 8.3977 (Å) is comparable with reported literature [1].



Fig.1 XRD pattern of Ni_{0.50}Zn_{0.50}Fe₂O₄

The crystallite size of the nanocrystalline sample for 311 planes was calculated using Debye-Scherrer formula given in equation (1).

where D is the crystallite size in nm, λ is the X-ray wavelength in Å, β is the line broadening at full width half maxima intensity (FWHM) in radians and θ is Bragg angle. The crystallize size was found to be 41 nm ± 5%.



Fig.2 TEM image of $Ni_{0.50}Zn_{0.50}Fe_2O_4$





The TEM image of the sample is as shown in Fig. 2. The particle sizes vary in the range of 15 nm to 64 nm. The AFM image of the sample in Fig. 3 show that particle sizes vary in the large range of 20 nm to 80 nm. The porosity of the sample was found to be $24\% \pm 5\%$.

3.2 IR analysis

The IR absorption spectrum of sample show two absorption bands as can be seen in Fig. 4. The higher band v_1 is between wave number 600–550 cm⁻¹ whereas lower band v_2 between 450–385 cm⁻¹. This is a common feature of all the ferrites indicating single phase spinel structure having two sub-lattices [4-7]. The highest band corresponds to intrinsic stretching vibrations of metals at the tetrahedral site, whereas lowest band is assigned to octahedral metal stretching.



Fig.4 IR spectrum of Ni_{0.50}Zn_{0.50}Fe₂O₄

3.3 DC resistivity

The plot of log ρ vs. 1000/T of the sample is shown in Fig. 5. It can be seen that the DC resistivity decreases with increasing temperature showing typical semiconductor like behavior. The variation in resistivity curve can be divided in to three temperature regions. Region (I) is in the relatively lower temperature range up to 400 K (~125°C). In this region resistivity does not exhibit much variation with temperature. The conduction phenomenon in this region is mostly attributed to scattering of charge carriers by lattice vibrations, impurities in the sample, presence of lattice defects (defects are having smaller mobility), voids and high resistive grain boundaries or existence of resonant phonon modes [8].



Fig.5 logp versus 1000/T of Ni_{0.50}Zn_{0.50}Fe₂O₄

The region (II) was observed in the temperature range around 400K to 675K $(125^{\circ}C \text{ to } 400^{\circ}C)$. In this region resistivity decreases with increase in the temperature and conduction attributed phenomenon is the to thermally activated mobility of the charge carriers. The change in slope of the curve was also found in this region. The small dip or cusps in the curves causing change in the slope can be clearly seen. This change is related to the magnetic transition from ferrimagnetic to paramagnetic state and corresponding temperature to the Curie point (T_c) of the sample [9]. In region (III) for temperature range T>673K (400° C), the resistivity decreases rapidly with temperature (above T_c). This decrease in resistivity is mainly associated to the increase in thermally activated drift mobility of the charge carriers in accordance with the hopping conduction model [10]. The higher temperature region above T_c is corresponding to a paramagnetic disordered region with comparatively high activation energy. The room temperature resistivity of nano samples is of the order of 6.5×10^7 ohm-cm.

3.4 M-H loop

The variation of magnetization, M, with the static applied magnetic field, H, (up to 2 T) at room temperature is as shown in Fig. 6. The magnetization increases with increasing applied magnetic field, attains its saturation value for fields higher than 0.50 T. The saturation magnetization was found to be 68 emu/g. The close hysteresis loop observed at room temperature indicates SPM nature of nano particles.



Fig.6 M-H loop of $Ni_{0.5}Zn_{0.5}Fe_2O_4$ at room temperature

3.5 Mössbauer spectrum

The room temperature Mössbauer spectrum recorded to monitor the local environment around Fe cations is as shown in Fig. 7. The solid line in the Fig.7 represents the simulated curves whereas solid circles represent experimental data points. Each spectrum was fitted with the three magnetic sextets which are arising due to Zeeman splitting. The two major sextets are corresponding to Fe ions residing at the A site (inner sextet) and B site (outer sextet) of the lattices. This is an indication of ferrimagnetic behavior of the samples [11]. The observed six-line magnetic patterns in the spectra are attributed to super exchange interaction between the magnetic ions at A and B sub-lattices.



Fig.7 Mössbauer spectrum of Ni_{0.50}Zn_{0.50}Fe₂O₄ The third broad featureless absorption sextet, characterized by a significant reduction of the magnetic hyperfine field is associated with relaxation effect. The Fe ions corresponding to broad sextet with much smaller hyperfine field than that of major sextet may be associated with the atoms near the surface of the

Table 1 The isomer shift (δ), quadrupole splitting (Δ), hyperfine Field values (H), outer line width (T) and areas in percentage of tetrahedral (A) and octahedral (B) sites occupied by Fe³⁺ ions of Ni_{0.50}Zn_{0.50}Fe₂O₄ sample derived from Mössbauer spectra recorded at room temperature. [Isomer shift values are relative to α -Fe (0.00 mm/s) foil]

Parameters	А	В	C	Doublet
Hyperfine				
Field	44.13	51.23	29.62	-
H (T)				
Isomer				
Shift	0.311	0.490	0.352	0.335
(δ) mm/s				
Quadrupole				
Splitting	0.003	0.111	0.018	0.433
Δ (mm/s)				
Outer				
Line width	2.240	0.185	6.556	0.423
T (mm/s)				
Area %	56.20	2.43	31.36	10.01

nano particles experiencing disordered spins [12]. This relaxation effect may also be due to the presence of strong inter-particle interactions and nano sizes of the particles of the samples [13]. The area of this broad sextet may be associated with the range of nano particles. The appearance of central doublet is indicating presence of superparamagnetic (SPM) particles in the sample. The various Mössbauer parameters of sample are given in Table 1. The results of the isomer shift vary in the range of 0.311 mm/s to 0.490 mm/s. This shows that, Fe ions are in the trivalent state. The quadrupole shift (Δ) values were found to be low, and is indicating that the local symmetry of the magnetic phase of A sites is close to cubic, while that of B site is close to trigonal [14, 15].

4. Conclusions

The nanoparticles of Ni_{0.5}Zn_{0.5}Fe₂O₄ were successfully synthesized by simple and easy cotton base wet chemical method. The crystallite size calculated using X-ray spectra is comparable with particle sizes obtained from TEM and AFM images. The high DC resistivity at room temperature is attributed to nano sizes of particles. The sample show high saturation magnetization and superparamagnetic nature at room temperature. The Fe ions at the A- and B- site are in trivalent state. The detailed dielectric studies can be undertaken in future.

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Email: manojkothawale@yahoo.com;

TABLES

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Captions for Figures

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- **Fig.2** TEM image of $Ni_{0.50}Zn_{0.50}Fe_2O_4$
- Fig.3 AFM image of Ni_{0.50}Zn_{0.50}Fe₂O₄
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